

## Supporting Information

### 2D MoO<sub>3-x</sub>S<sub>x</sub>/MoS<sub>2</sub> van der Waals Assembly:

### a Tunable Heterojunction with Attractive Properties for Photocatalysis

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## S1. Methodology and Computational Setup

### S1.1. Total energy calculations and electronic properties

Total energy calculations including geometry optimizations of all pristine and substituted single- and multi-layer structures were performed by the *ab initio* CRYSTAL17 code which works within periodic boundary conditions and adopts localized Gaussian-type function basis sets (BSs). The following BSs were used: Mo\_SC\_HAYWSC-311G(d31)\_cora\_1997<sup>1</sup> (for Mo atoms), S\_86-311G\*\_lichenot\_1993<sup>2</sup> (for S atoms) and O\_8-411d11G\_valenzano\_2006<sup>3</sup> (for O atoms). For the exchange-correlation potential, the PBE functional<sup>4</sup> have been adopted and the van der Waals contributions were described by using the semi-empirical Grimme D3 approach<sup>5</sup> with optimized scaling factors<sup>6</sup>. The tolerances for the evaluation of Coulomb and exchange series are set to  $8 \times 8 \times 8 \times 8 \times 16$ .<sup>7</sup> Since ordinary DFT within the PBE/GGA underestimate the band gap, all the electronic properties were computed via single-point calculations by using the range separated hybrid Heyd–Scuseria–Ernzerhof (HSE06) exchange correlation functional<sup>8</sup> on the PBE optimized geometries using CRYSTAL17. All calculations (including geometry optimizations and electronic properties) were performed with an energy threshold of  $10^{-10}$  Ha per unit-cell for the self-consistent field (SCF) process. The reciprocal space for the unit-cell of pristine 2D MoS<sub>2</sub> and MoO<sub>3</sub> systems is sampled according to a sublattice with a 12×12 k-point mesh. The k-point mesh sampling is progressively reduced as the size of the monolayer and multilayer unit cells increases for the doped materials (see Tables S3 and S4).

Three different energetic stability analysis were performed to analyze the stability of 2D MoO<sub>3-x</sub>S<sub>x</sub> and MoS<sub>2-x</sub>O<sub>x</sub> systems: (i) binding energy, (ii) reaction energy and (iii) thermodynamic phase diagram which were described in our previous work of the same systems in the bulk phase.<sup>6</sup>

### S1.2. Optical properties

The optical properties in this work were calculated through the frequency dependent dielectric matrix  $\epsilon_{\alpha\beta}(\omega)$  within the HSE06 functional by using the Vienna *ab initio* simulation package (VASP).<sup>9-11</sup> The previously PBE optimized structures with CRYSTAL were used for the optical calculations. Electronic wave functions have been expanded using a plane wave basis set with a cutoff energy of 500 eV for the valence electrons and the interaction between the valence and the core electrons for Mo, O and S atoms was described with the projected augmented wave (PAW) method.<sup>12</sup> In all optical calculations a cell with a vacuum region of 20 Å in the direction normal to the sheet plane has been used to avoid image-image interactions. The number of empty conduction bands was converged for each structure with respect to standard calculations. The absorption coefficient can be evaluated from dielectric matrix using the Kramers-Kronig transformation:<sup>13</sup>

$$\alpha_{\alpha\beta}(\omega) = \frac{2\omega k_{\alpha\beta}(\omega)}{c} = \frac{\omega \text{Im}(\epsilon_{\alpha\beta}(\omega))}{c n_{\alpha\beta}(\omega)} \quad (1)$$

where  $c$  is the speed of light in vacuum.  $n_{\alpha\beta}$  and  $k_{\alpha\beta}$  are real and imaginary parts of the complex refractive index, and are known as the refractive index and the extinction index, respectively. They are given by the following relations:

$$n_{\alpha\beta}(\omega) = \sqrt{\frac{|\epsilon_{\alpha\beta}(\omega)| + \text{Re}(\epsilon_{\alpha\beta}(\omega))}{2}} \quad (2) \quad k_{\alpha\beta}(\omega) = \sqrt{\frac{|\epsilon_{\alpha\beta}(\omega)| - \text{Re}(\epsilon_{\alpha\beta}(\omega))}{2}} \quad (3)$$

### S1.3. High frequency dielectric constant profile $\epsilon_{\infty}(\mathbf{z})$

To obtain the local (microscopic) dielectric constant profiles, we followed a DFT scheme based on polarization variation in response to a finite electric field as recently proposed<sup>14-16</sup> to extract dielectric profiles in heterostructures. This method avoids the complete determination of the dielectric matrix, thus reducing the computational time.

In order to extract the optical (high frequency) dielectric constant profile along the out-of-plane direction ( $z$ -axis) in 2D systems, the calculated DFT electron density are analyzed in four steps. In the first step, the calculated electronic density with ( $\rho(z)|_{\vec{E}_{ext}}$ ) and without ( $\rho(z)|_0$ ) external electric field using CRYSTAL package is averaged in the plane parallel to the  $z$ -axis. It is then smoothed using a nanoscale averaging<sup>17</sup> along the out-of-plane direction. It is noteworthy that for the high frequency limit (optical dielectric constant) only the electrons are considered to respond to the applied electric field whereas for lower frequencies (static dielectric constant) both electronic and ionic (phonons) effects are considered. In the second step, the induced electron density,  $\delta\rho_{ind}(z)$ , is computed by subtracting the results of electronic density with and without external electric field:

$$\delta\rho_{ind}(z) = \rho(z)|_{\vec{E}_{ext}} - \rho(z)|_0 \quad (4)$$

The third step is computing the induced polarization,  $p_{ind}(z)$ , by the partial integration of the induced electronic density:

$$\frac{dp_{ind}(z)}{dz} = -\delta\rho(z) \quad (5)$$

In the fourth step, the nanoscopic average of the high frequency dielectric constant profile  $\epsilon_\infty(z)$  is calculated from the induced polarization:

$$\epsilon_\infty(z) = \frac{\epsilon_0 E_{ext}}{\epsilon_0 E_{ext} - p_{ind}(z)} \quad (6)$$

We can finally define the inverse of effective dielectric constant ( $\epsilon_{eff}$ ) of a 2D system (of height  $x_1$ - $x_2$ ) as the average of the inverse of local dielectric function:<sup>18</sup>

$$\frac{1}{\epsilon_{eff}} = \frac{1}{x_2 - x_1} \int_{x_1}^{x_2} \frac{1}{\epsilon_\infty(x)} dx \quad (7)$$

This method has been performed<sup>15,16,19-21</sup> to calculate dielectric constant profile using the local orbital SIESTA<sup>22</sup> code so far. For the first time we performed this method using the CRYSTAL results of electronic density. It is worth to note that the calculations presented in all previous published papers were performed using the simple LDA<sup>23</sup> or GGA<sup>4</sup> functional while in our current work we employed hybrid functional of HSE06<sup>8</sup> to provide more accurate estimations.

## S2. Pristine structures of single- and few-layers MoO<sub>3</sub> and MoS<sub>2</sub>

### S2.1. Structures, binding energy and exfoliation energy

In order to model the MoO<sub>3</sub> and MoS<sub>2</sub> nanosheet structures, the primitive cell of 2D layers was directly cut from the optimized 3D  $\alpha$ -MoO<sub>3</sub> and 2H-MoS<sub>2</sub> bulk structures along the direction of the vdW interaction. The crystal structures of single-layer (SL) and few-layer (FL) MoO<sub>3</sub> and MoS<sub>2</sub> systems is depicted in Figure S1a and b. In each layer of MoO<sub>3</sub> system, the orthorhombic lattice consists of 2 Mo and 6 O atoms where Mo atoms are an octahedral environment connected by covalent bonds to 1 O-oxo species and 5 bridging O atoms. The atomic arrangement in each layer of MoS<sub>2</sub> system (Figure S1b) is hexagonal, in which each Mo atom is surrounded at equal distances by 6 S atoms placed at the corners of a trigonal prism.

To evaluate the stability of pristine single- and few-layer MoO<sub>3</sub> structures, the binding energies ( $E_b$ ) per MoO<sub>3</sub> were computed as

$$E_b = \left( \frac{E_{tot} - (n_{Mo}E_{Mo} + (\frac{1}{2})n_OE_{O_2})}{N} \right) \times 4 \quad (8)$$

and the binding energies of pristine single- and few-layer MoS<sub>2</sub> structures per MoS<sub>2</sub> were computed as

$$E_b = \left( \frac{E_{tot} - (n_{Mo}E_{Mo} + n_S E_{S_2})}{N} \right) \times 3 \quad (9)$$

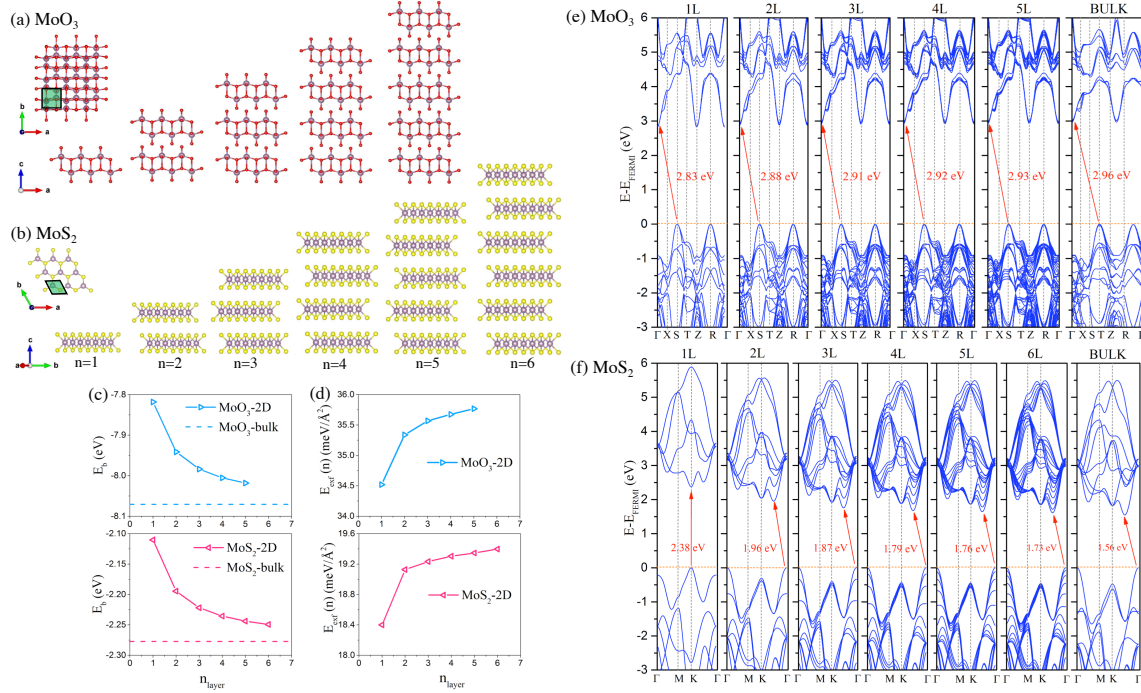
where  $E_{tot}$  is the total energy per cell,  $E_i$  is the energy of the  $i$ -th individual elements in their respective ground states,  $n_i$  is the number of species  $i$  in the structure and  $N$  is the total number of atoms per cell.

The  $n$ -layer exfoliation energy per unit area  $E_{exf}(n)$  is given by<sup>24</sup>

$$E_{exf}(n) = \frac{E_{iso}(n) - E_{bulk}n/m}{\mathcal{A}} \quad (10)$$

where  $E_{iso}(n)$  is the energy of the unit cell of an isolated  $n$ -layer slab in vacuum,  $E_{bulk}$  is the energy of the unit cell of a bulk material composed of  $m$  layers, thus  $E_{bulk}/m$  corresponds to the energy of the bulk per layer, and  $\mathcal{A}$  is the in-plane area of the bulk unit cell.

According to our results, the lattice parameters and exfoliation energies of both MoO<sub>3</sub> and MoS<sub>2</sub> systems slightly increases as the number of layers is increased from monolayer to bulk, while the binding energy is decreased (see Figure S1, Tables S1 and S2).



**Figure S1.** Top and side view of  $(3 \times 3)$  supercell crystal structures (a, b), calculated layer depended binding energy (c), exfoliation energy (d) and electronic band structure (e, f) of the 1L-5L MoO<sub>3</sub> and 1L-6L MoS<sub>2</sub> systems. The purple, red and yellow balls in the geometrical models represent the Mo, O and S atoms, respectively. The binding energy and electronic band structure for 3D bulk phases are also presented.

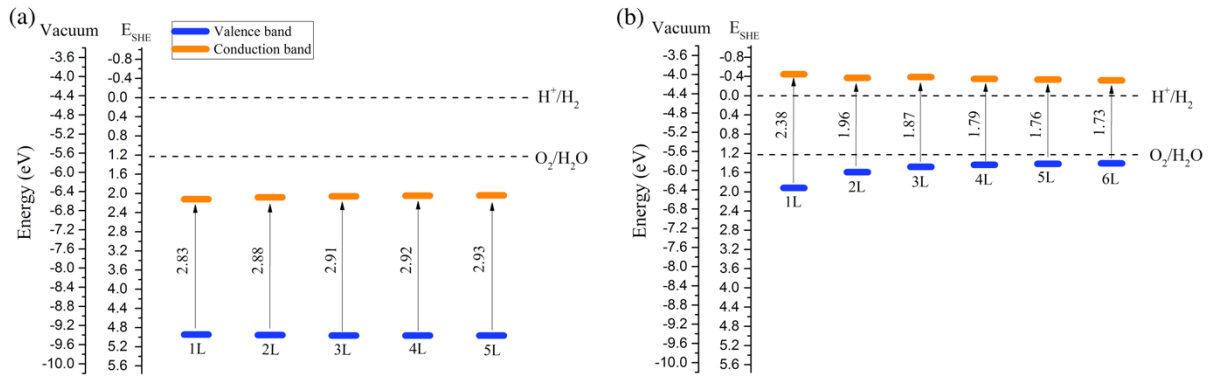
MoO <sub>3</sub>	$a$ (Å)	$b$ (Å)	$d$ (Å)	$T$ (Å)	$E_b$ (eV)	$E_{exf}$ (meV/Å <sup>2</sup> )	$E_g$ (eV)	VB (eV)	CB (eV)	$\epsilon_{eff}$
1L	3.706	3.910	---	6.401	-7.82	34.52	2.83	-9.39	-6.56	2.85
2L	3.710	3.927	0.601	6.391	-7.94	35.34	2.88	-9.4	-6.52	3.35
3L	3.711	3.930	0.590	6.388	-7.98	35.57	2.91	-9.41	-6.49	3.51
4L	3.712	3.934	0.585	6.386	-8.00	35.67	2.92	-9.41	-6.48	3.60
5L	3.713	3.936	0.584	6.385	-8.02	35.77	2.93	-9.41	-6.44	3.68
Bulk	3.72 (3.70) <sup>25</sup>	3.94 (3.96) <sup>25</sup>	0.580	6.370	-8.07 (-7.80) <sup>26</sup>	---	2.96 (3.0) <sup>27</sup>	---	---	4.3

**Table S1.** Calculated optimized lattice parameters ( $a$  and  $b$ ), the average minimum distance between the O<sub>T</sub>-planes of adjacent layers ( $d$ ), the average thickness of each layer ( $T$ ), binding energy ( $E_b$ ) per MoO<sub>3</sub>, the exfoliation energy ( $E_{exf}$ ), the HSE06 electronic band gap ( $E_g$ ), absolute valence and conduction band-edge positions (VB and CB) and effective high frequency dielectric constant ( $\epsilon_{eff}$ ) along out-of-plane direction of pristine 1L-5L MoO<sub>3</sub> structures. The simulated results for 3D bulk phase of  $\alpha$ -MoO<sub>3</sub> are also presented. The values in parenthesis are the corresponding experimental values<sup>25-27</sup> (the simulated lattice parameter for MoO<sub>3</sub> bulk along the direction of the vdW interaction is 13.90 Å, agree well with the experimental data 13.87 Å<sup>25</sup>).

2D-MoS <sub>2</sub>	<i>a</i> (Å)	<i>d</i> (Å)	<i>T</i> (Å)	<i>E<sub>b</sub></i> (eV)	<i>E<sub>exf</sub></i> (meV/Å <sup>2</sup> )	<i>E<sub>g</sub></i> (eV)	<i>VB</i> (eV)	<i>CB</i> (eV)	<i>ε<sub>eff</sub></i>
1L	3.154 (3.16) <sup>28</sup>	---	3.165	-2.110	18.40	2.38 (2.40) <sup>29</sup>	-6.36 (-7.46) <sup>30</sup>	-3.99 (-4.7) <sup>30</sup>	4.96
2L	3.155	2.971	3.166	-2.194	19.128	1.96 (2.10) <sup>29</sup>	-6.03 (-6.3) <sup>30</sup>	-4.07 (-4.4) <sup>30</sup>	5.86
3L	3.155	2.968	3.167	-2.222	19.233	1.87 (1.75) <sup>29</sup>	-5.92 (-6.0) <sup>30</sup>	-4.05 (-4.3) <sup>30</sup>	6.13
4L	3.156	2.966	3.167	-2.236	19.304	1.79	-5.88 (-5.6) <sup>30</sup>	-4.09 (-4.0) <sup>30</sup>	6.23
5L	3.156	2.964	3.168	-2.244	19.346	1.76	-5.86	-4.10	6.26
6L	3.156	2.962	3.169	-2.250	19.398	1.73	-5.85	-4.12	6.26
Bulk	3.16 (3.16) <sup>31</sup>	2.94	3.17	-2.28 (-2.81) <sup>32</sup>	---	1.56 (1.23) <sup>33</sup>	---	---	6.30

**Table S2.** Optimized lattice parameters (*a*), the average minimum distance between the S-planes of adjacent layers (*d*), the average thickness of each layer (*T*), binding energy (*E<sub>b</sub>*) per MoS<sub>2</sub>, the exfoliation energy (*E<sub>exf</sub>*), the HSE06 electronic band gap (*E<sub>g</sub>*), absolute valence and conduction band-edge position (VB and CB) and effective high frequency dielectric constant (*ε<sub>eff</sub>*) along out-of-plane direction of pristine 1L-6L MoS<sub>2</sub> structures. The simulated results for 3D bulk phase of 2H-MoS<sub>2</sub> are also presented. The simulated lattice parameter for 2H-MoS<sub>2</sub> bulk along the direction of the vdW interaction is 12.28 Å, agree well with the experimental data 12.29 Å<sup>31</sup>. Note that the experimental band gap values are optical band gap.

## S2.2. Absolute valence and conduction band position



**Figure S2.** Calculated conduction and valence band-edge positions for 1L-5L MoO<sub>3</sub> (a) and 1L-6L MoS<sub>2</sub> (b) respect to vacuum level and standard hydrogen electrode. The lower edge of the conduction band (orange colour) and upper edge of the valence band (blue colour) are presented along with the band gap in electron volts. The dashed black lines indicate the water stability limits for hydrogen and oxygen evolution. The absolute potential of the standard hydrogen electrode was taken as 4.44 eV at a pH = 0.

### S3. MoO<sub>3</sub>/MoS<sub>2</sub> interface

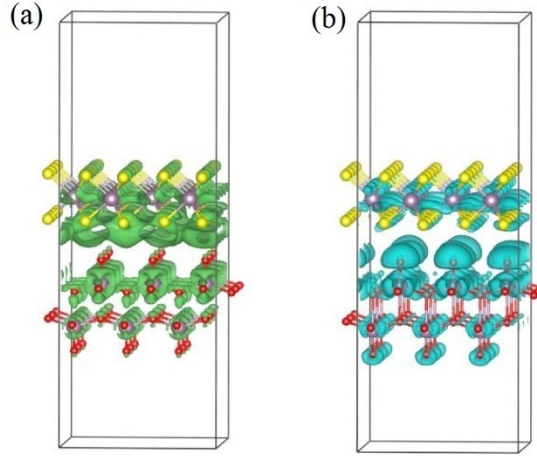


Fig. S3. Charge density difference  $\Delta\rho = \rho(\text{MoO}_3/\text{MoS}_2) - \rho(\text{MoO}_3) - \rho(\text{MoS}_2)$  for MoO<sub>3</sub>/MoS<sub>2</sub> heterojunction.  $\Delta\rho$  is shown by isosurfaces bounding regions of electron accumulation at  $+1.2 \times 10^{-4} \text{ e}/\text{\AA}^3$  (a) and electron depletion at  $-1.2 \times 10^{-4} \text{ e}/\text{\AA}^3$  (b).

### S4. S-doped 1L MoO<sub>3</sub> and O doped 1L MoS<sub>2</sub>

#### S4.1. Structures and binding energies

To study the effects of sulfur doped SL MoO<sub>3</sub> and oxygen doped SL MoS<sub>2</sub> structures, different dopant concentrations were considered for both systems. To this aim, a host unit-cell structure of aforementioned systems was adopted for high dopant concentration levels while different size supercells (3×3) and (2×2) were used for low concentration levels. For S-doped SL MoO<sub>3</sub> and O-doped SL MoS<sub>2</sub>, the maximum doping concentration is considered up to 33% and 100%, respectively. For each doping level, several configurations were tested and only the properties of the most stable ones are presented in this work.

To evaluate the stability of these structures, the binding energies per MoO<sub>3</sub> of S-doped SL MoO<sub>3</sub> supercell were computed as:

$$E_b = \left( \frac{E_{tot} - (n_{Mo}E_{Mo} + (\frac{1}{2})n_OE_{O_2} + n_S E_{S\alpha})}{N} \right) \times 4 \quad (11)$$

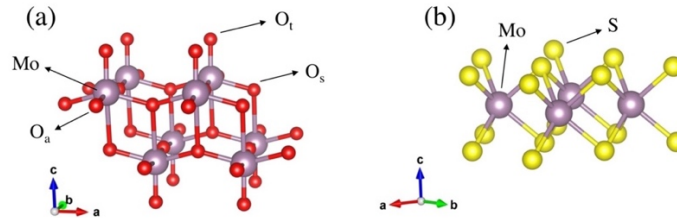
and the binding energies of O-doped SL MoS<sub>2</sub> structures per MoS<sub>2</sub> were computed as

$$E_b = \left( \frac{E_{tot} - (n_{Mo}E_{Mo} + n_S E_{S\alpha} + (\frac{1}{2})n_O E_{O_2})}{N} \right) \times 3 \quad (12)$$

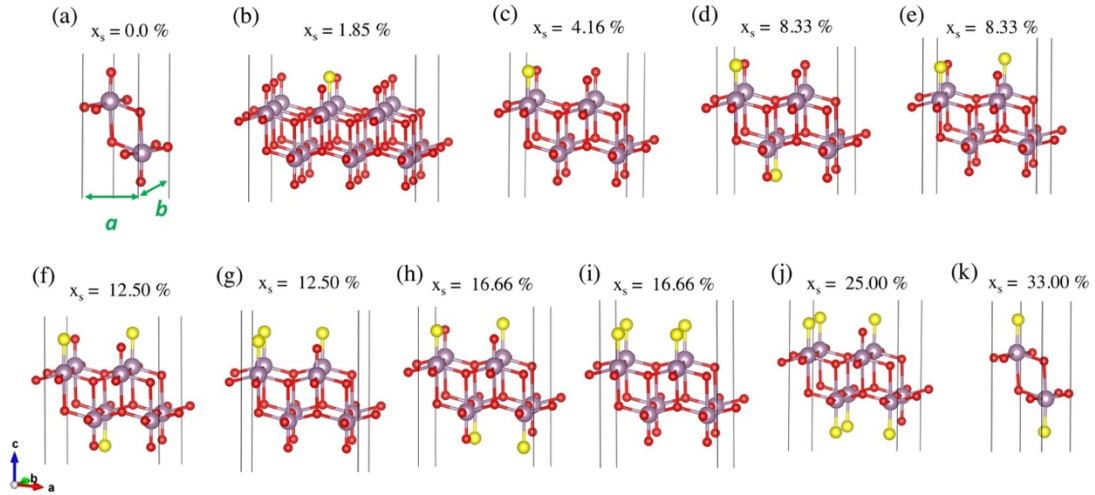
where  $E_{tot}$  is the total energy per cell,  $E_i$  is the energy of the  $i$ -th individual elements in their respective ground states,  $n_i$  is the number of species  $i$  in the structure and  $N$  is the total number of atoms per cell. We used the conventional reference states, solid molybdenum, solid sulfur ( $\alpha$ -sulfur) and isolated O<sub>2</sub> to compute the binding energies.

S concentration (%)	System	corresponding cell	corresponding figure	<i>k</i> -point mesh	<i>a</i> (Å)	<i>b</i> (Å)	<i>E<sub>b</sub></i> (eV)	<i>E<sub>g</sub></i> (eV)	$\epsilon_{eff}$
0.0	MoO <sub>3</sub>	unitcell	Fig. S6 (a)	12×12×1	3.706	3.910	-7.82	2.83	2.85
1.85	MoS <sub>0.06</sub> O <sub>2.94</sub>	(3 × 3)	Fig. S6 (b)	4×4×1	3.705	3.914	-7.70	2.40	2.93
4.16	MoS <sub>0.12</sub> O <sub>2.88</sub>	(2 × 2)	Fig. S6 (c)	6×6×1	3.705	3.915	-7.55	2.41	3.06
8.33	MoS <sub>0.25</sub> O <sub>2.75</sub>	(2 × 2)	Fig. S6 (d)	6×6×1	3.704	3.913	-7.29	2.44	3.24
8.33	MoS <sub>0.25</sub> O <sub>2.75</sub>	(2 × 2)	Fig. S6 (e)	6×6×1	3.704	3.913	-7.29	2.41	3.33
12.5	MoS <sub>0.38</sub> O <sub>2.62</sub>	(2 × 2)	Fig. S6 (f)	6×6×1	3.702	3.910	-7.02	2.42	3.52
12.5	MoS <sub>0.38</sub> O <sub>2.62</sub>	(2 × 2)	Fig. S6 (g)	6×6×1	3.710	3.905	-7.04	1.98	3.65
16.66	MoS <sub>0.5</sub> O <sub>2.5</sub>	(2 × 2)	Fig. S6 (h)	6×6×1	3.703	3.910	-6.76	2.48	3.79
16.66	MoS <sub>0.5</sub> O <sub>2.5</sub>	(2 × 2)	Fig. S6 (i)	6×6×1	3.715	3.910	-6.78	2.06	3.99
25	MoS <sub>0.75</sub> O <sub>2.25</sub>	(2 × 2)	Fig. S6 (j)	6×6×1	3.711	3.90	-6.25	2.41	4.16
33	MoSO <sub>2</sub>	unitcell	Fig. S6 (k)	12×12×1	3.72	3.89	-5.75	2.51	4.49

**Table S3.** Different models, corresponding figures, *k*-point mesh, calculated optimized lattice parameters (*a* and *b*), binding energy (*E<sub>b</sub>*) per MoO<sub>3</sub>, the electronic band gap (*E<sub>g</sub>*) and effective high frequency dielectric constant ( $\epsilon_{eff}$ ) along out-of-plane direction for the S-substituted SL MoO<sub>3</sub> for different concentrations.



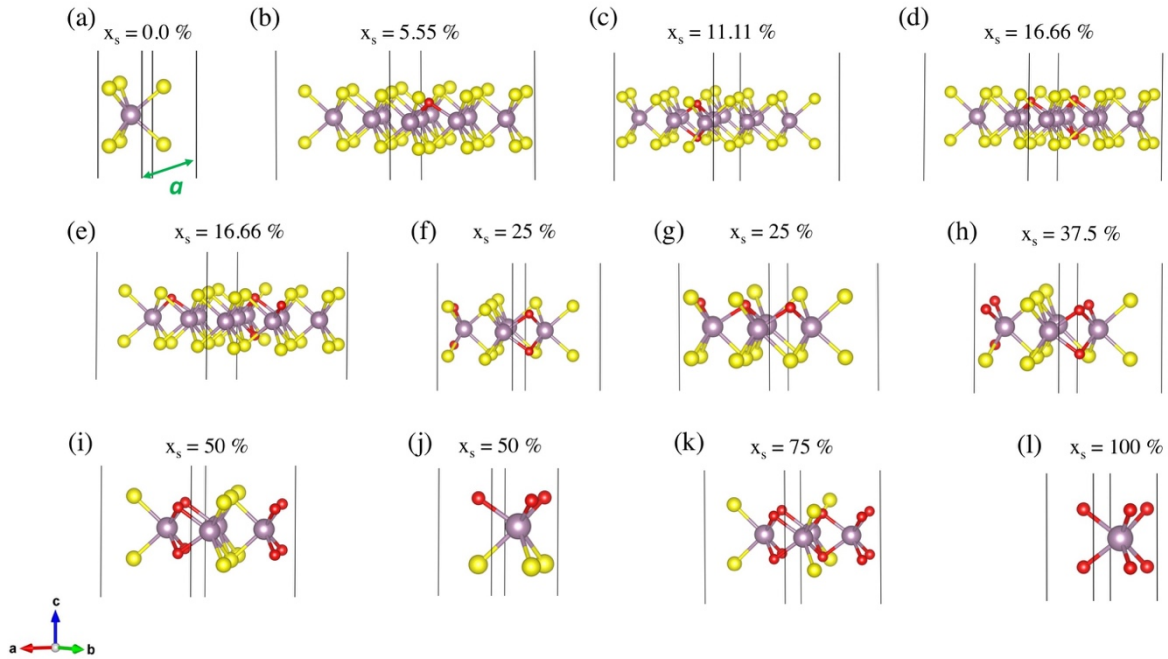
**Figure S4.** The (2×2) super-cells of (a) SL MoO<sub>3</sub> and (b) SL MoS<sub>2</sub>. The unit-cell of SL MoO<sub>3</sub> includes 2Mo, 2 O<sub>t</sub> (the apical oxygen atoms), 2 O<sub>a</sub> (the corner-sharing oxygen atoms) and 2 O<sub>s</sub> (the edge-sharing oxygen atoms), while the unit-cell of SL MoS<sub>2</sub> includes 1 Mo and 2 equivalent S. Color code as in Figure S1.



**Figure S5.** Optimized structures of S-doped SL MoO<sub>3</sub> for different concentrations (with increasing S content when going from (a) to (k)). Color code as in Figure S1. The lattice parameters of pristine structure are also shown with green arrows.

O concentration (%)	System	corresponding cell	corresponding figure	$k$ -point mesh	$a$ (Å)	$E_b$ (eV)	$E_g$ (eV)	$\epsilon_{eff}$
0.0	MoS <sub>2</sub>	unitcell	Fig. S7 (a)	12×12×1	3.155	-2.11	2.38	4.96
5.55	MoO <sub>0.11</sub> S <sub>1.89</sub>	(3 × 3)	Fig. S7 (b)	4×4×1	3.130	-2.26	2.41	4.67
11.11	MoO <sub>0.22</sub> S <sub>1.78</sub>	(3 × 3)	Fig. S7 (c)	4×4×1	3.115	-2.43	2.47	4.58
16.66	MoO <sub>0.33</sub> S <sub>1.67</sub>	(3 × 3)	Fig. S7 (d)	4×4×1	3.096	-2.58	2.39	4.42
16.66	MoO <sub>0.33</sub> S <sub>1.67</sub>	(3 × 3)	Fig. S7 (e)	4×4×1	3.093	-2.58	2.42	4.41
25	MoO <sub>0.5</sub> S <sub>1.5</sub>	(2 × 2)	Fig. S7 (f)	6×6×1	3.065	-2.82	2.25	4.68
25	MoO <sub>0.5</sub> S <sub>1.5</sub>	(2 × 2)	Fig. S7 (g)	6×6×1	3.056	-2.76	2.12	4.40
37.5	MoO <sub>0.75</sub> S <sub>1.25</sub>	(2 × 2)	Fig. S7 (h)	6×6×1	3.010	-3.17	2.19	4.42
50	MoOS	(2 × 2)	Fig. S7 (i)	6×6×1	2.960	-3.59	2.13	3.73
50	MoOS	unitcell	Fig. S7 (j)	12×12×1	2.987	-3.37	1.79	3.34
75	MoO <sub>1.5</sub> S <sub>0.5</sub>	(2 × 2)	Fig. S7 (k)	6×6×1	2.891	-4.40	2.00	3.07
100	MoO <sub>2</sub>	unitcell	Fig. S7 (l)	12×12×1	2.818	-5.24	1.63	2.34

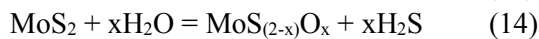
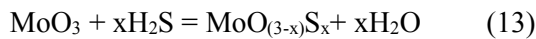
**Table S4.** Different models, corresponding figures,  $k$ -point mesh, calculated optimized lattice parameter ( $a$ ), binding energy ( $E_b$ ) per MoS<sub>2</sub>, the electronic band gap ( $E_g$ ) and effective high frequency dielectric constant ( $\epsilon_{eff}$ ) along out-of-plane direction for the O-substituted SL MoS<sub>2</sub> for different concentrations.



**Figure S6.** Optimized structures of O-doped SL MoS<sub>2</sub> for different concentrations (with increasing oxygen concentration when going from (a) to (l)). Color code as in Figure S1. The lattice parameters of pristine structure are also shown with green arrows.

## S4.2. Thermodynamic analysis

For the more discerning of the stability of  $SLMoO_{3-x}S_x$  and  $MoS_{2-x}O_x$  compositions as a function of the number of S- or O-atoms exchanged in the MoO<sub>3</sub> and MoS<sub>2</sub> single-layer structures respectively, we calculate the Grand potential,  $\Omega$ , assuming that the reservoir surrounding the solids is constituted from ideal gas phase mixtures of H<sub>2</sub>S/H<sub>2</sub>O (commonly used experimentally as sulfiding and oxidizing agents):





$$\Omega_{\text{MoO}_{3-x}\text{S}_x} = E_b(\text{MoO}_{3-x}\text{S}_x) - E_b(\text{MoO}_3) + xG_{f,T_0}^0(\text{H}_2\text{O}) - xG_{f,T_0}^0(\text{H}_2\text{S}) - xRT_0 \ln \left[ \frac{p(\text{H}_2\text{S})}{p(\text{H}_2\text{O})} \right] \quad (15)$$

$$\Omega_{\text{MoS}_{2-x}\text{O}_x} = E_b(\text{MoS}_{2-x}\text{O}_x) - E_b(\text{MoS}_2) + xG_{f,T_0}^0(\text{H}_2\text{S}) - xG_{f,T_0}^0(\text{H}_2\text{O}) - xRT_0 \ln \left[ \frac{p(\text{H}_2\text{O})}{p(\text{H}_2\text{S})} \right] \quad (16)$$

where  $E_b$  stands for the 0 K, binding energies of the different solids involved (neglecting vibrational and entropic contributions, see also supporting information),  $G_{f,T_0}^0$  stands for the Gibbs free energies of formation of  $\text{H}_2\text{S}$  and  $\text{H}_2\text{O}$  molecules (including thermal and entropic effects evaluated from NIST data base at  $T_0=298$  K). So, for each value of  $x$ , we will plot the evolution of  $\Omega_x$  as a function of the variable  $\pm RT_0 \ln \left[ \frac{p(\text{H}_2\text{O})}{p(\text{H}_2\text{S})} \right]$  fixing the partial pressures of  $\text{H}_2\text{S}/\text{H}_2\text{O}$  in the reservoir assumed at  $T_0=298$  K. It is worth to note that increasing  $T$  slightly diminishes (increases, respectively) the free enthalpy of oxidation (sulfidation, respectively) reaction (supplementary information). However, the main trends reported at ambient  $T$  are only very weakly affected.

Fig. S8 a and b illustrates the thermodynamic phase diagrams of substituted single-layer systems with respect to the pristine structures considering the  $\text{H}_2\text{S}/\text{H}_2\text{O}$  reservoir by using the Grand potential approach. It can be seen that for a  $RT_0 \ln \left[ \frac{p(\text{H}_2\text{S})}{p(\text{H}_2\text{O})} \right]$  term greater than  $\sim -0.27$  eV, the pristine structure of SL  $\text{MoO}_3$  becomes sulfided. For  $RT_0 \ln \left[ \frac{p(\text{H}_2\text{S})}{p(\text{H}_2\text{O})} \right]$  in the range between  $-0.27$  and  $-0.19$  eV, the structure with 4.16% S is the most stable. For  $RT_0 \ln \left[ \frac{p(\text{H}_2\text{S})}{p(\text{H}_2\text{O})} \right]$  in the range between  $-0.19$  and  $-0.16$  eV, the structure with 16.7% S is the most stable. For  $RT_0 \ln \left[ \frac{p(\text{H}_2\text{S})}{p(\text{H}_2\text{O})} \right]$  higher than  $-0.16$  eV, system with 33% S is thermodynamically favored. By contrast, the oxidation of SL  $\text{MoS}_2$  requires value of  $RT_0 \ln \left[ \frac{p(\text{H}_2\text{O})}{p(\text{H}_2\text{S})} \right]$  as high as  $\sim 0.66$  eV, which implies that the SL  $\text{MoS}_2$  pristine system should remain stable while the structure with the highest O concentration (100%) should be favored.

Moreover, as an alternative pathway for the oxidation of SL  $\text{MoS}_2$ , we can also use  $\text{O}_2/\text{SO}_2$ :



Neglecting the entropic changes for the solid phases, the Grand potential can be approximated as follows:

$$\Omega_{\text{MoS}_{2-x}\text{O}_x} = E_b(\text{MoS}_{2-x}\text{O}_x) - E_b(\text{MoS}_2) + xG_{f,T_0}^0(\text{SO}_2) - \frac{3}{2xG_{f,T_0}^0(\text{O}_2)} - xRT_0 \ln \left[ \frac{p(\text{O}_2)^{\frac{3}{2}}}{p(\text{SO}_2)p_0^{\frac{1}{2}}} \right] \quad (18)$$

It can be easily shown that by using  $\text{O}_2$  as oxidizer and  $\text{SO}_2$  as a byproduct, reaction energies become exothermic and the thermodynamic diagram can be shifted to accessible reaction conditions (Fig. S8 c) which reveals that the chemical potential of  $\text{O}_2$  required for oxidizing SL  $\text{MoS}_2$  is very low in comparison to  $\text{H}_2\text{O}$ . This also shows that the full conversion of SL  $\text{MoS}_2$  into SL  $\text{MoO}_3$  is strongly exothermic process, which makes more difficult the control of the partial sulfidation state under  $\text{O}_2/\text{SO}_2$  environment than under  $\text{H}_2\text{O}/\text{H}_2\text{S}$ .

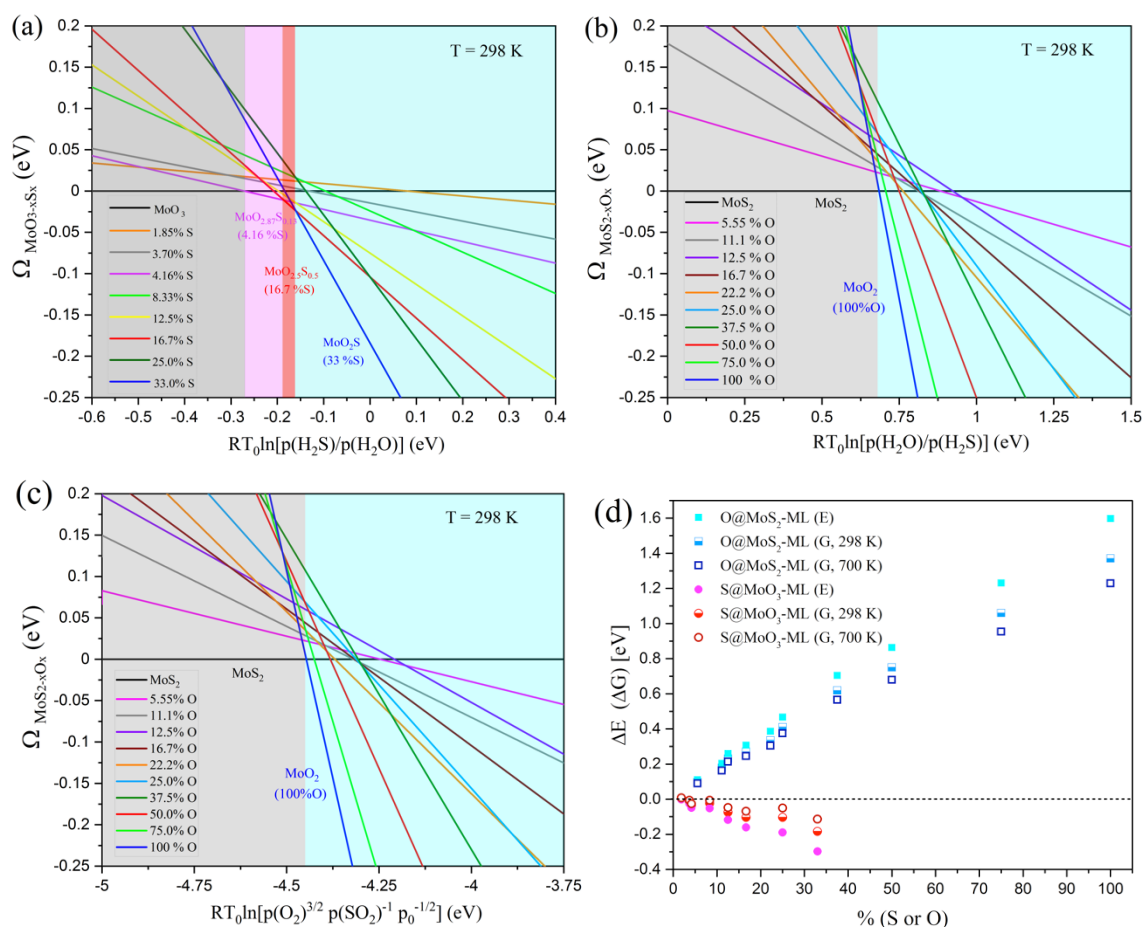
By using equations S13 and S14 we can calculate the reaction energies for the most stable S-substituted SL  $\text{MoO}_3$  and O-substituted SL  $\text{MoS}_2$  nanostructures for different concentrations as depicted in Fig S8 d. In this figure,  $\Delta E$  is the difference of 0K binding energy of the S-substituted SL  $\text{MoO}_3$  and O-substituted SL  $\text{MoS}_2$  systems with respect to DFT binding energies of  $\text{H}_2\text{S}$  and  $\text{H}_2\text{O}$  as reactants/products:

$$\Delta E = E_b(\text{MoO}_{3-x}\text{S}_x) - E_b(\text{MoO}_3) + xE_b(\text{H}_2\text{O}) - xE_b(\text{H}_2\text{S}) \quad (19)$$

$\Delta G$  is the difference of binding energy of the S-substituted SL  $\text{MoO}_3$  and O-substituted SL  $\text{MoS}_2$  systems with respect to experimental free energies (enthalpy and entropy) of  $\text{H}_2\text{S}$  and  $\text{H}_2\text{O}$  given by NIST thermodynamic database:

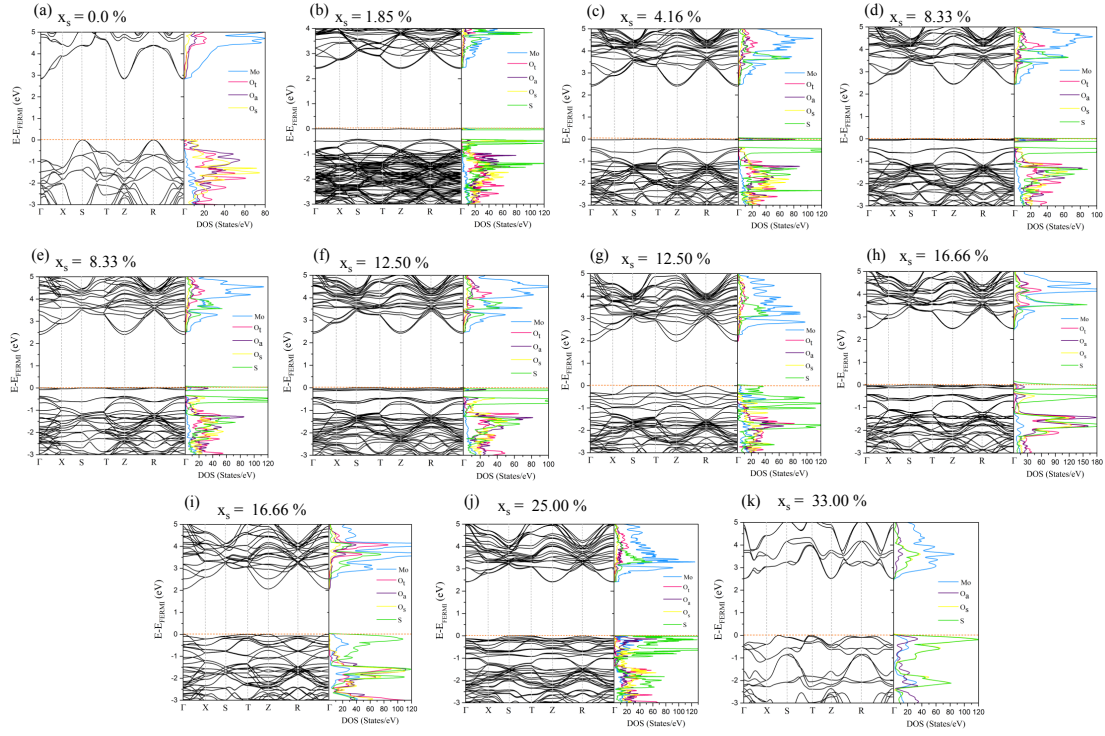
$$\Delta G = E_b(\text{MoO}_{3-x}\text{S}_x) - E_b(\text{MoO}_3) + xG_b(\text{H}_2\text{O}) - xG_b(\text{H}_2\text{S}) \quad (20)$$

The reaction energies for the oxidation of ML  $\text{MoS}_2$  are continuously increasing as a function of  $x$ , while the sulfidation energies of SL  $\text{MoO}_3$  are continuously decreasing that reveals that the sulfidation process of SL  $\text{MoO}_3$  is thermodynamically more favorable than the oxidation one SL  $\text{MoS}_2$  when we consider  $\text{H}_2\text{S}/\text{H}_2\text{O}$  as reactants or products.

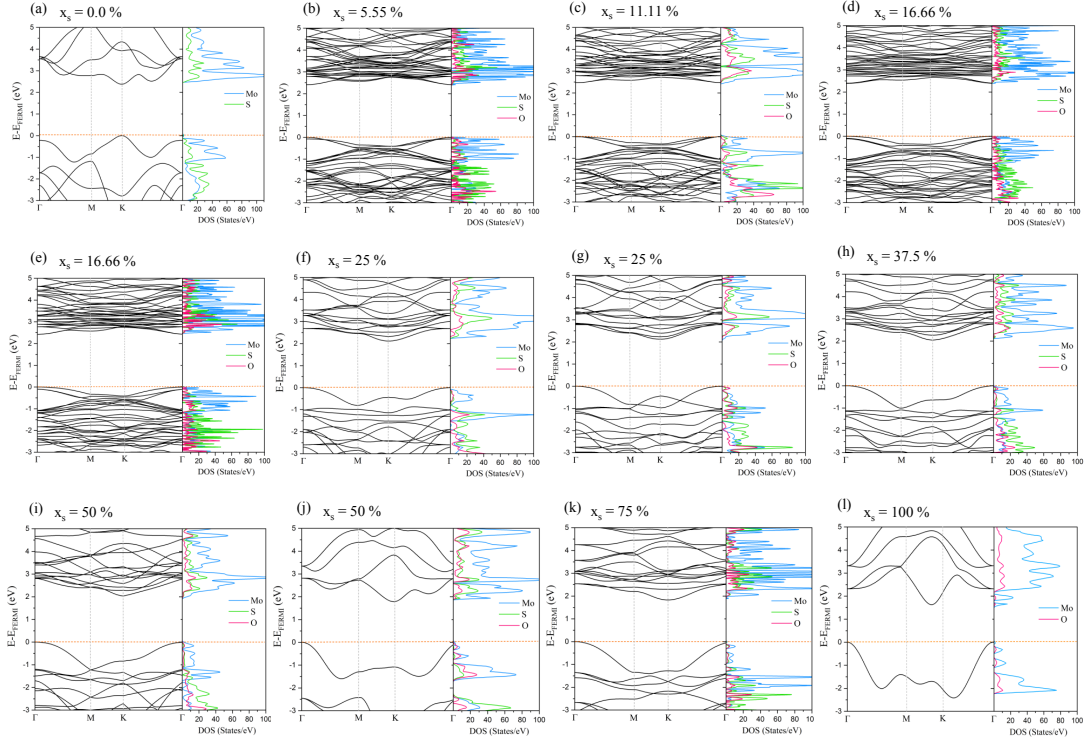


**Figure S7.** Thermodynamic phase stability of the S-substituted SL  $\text{MoO}_3$  (a) and O-substituted SL  $\text{MoS}_2$  (b) structures for different S- and O-concentrations with respect to pristine structures considering the  $\text{H}_2\text{S}/\text{H}_2\text{O}$  reservoir. (c) Thermodynamic phase stability of the O-substituted SL  $\text{MoS}_2$  structures for different O-concentrations with respect to pristine structure considering  $\text{O}_2$  reservoir and formation of solid sulfur. (d) The calculated reaction energy for the most stable S-substituted SL  $\text{MoO}_3$  and O-substituted SL  $\text{MoS}_2$  structures for different concentrations.

### S4.3. Electronic and optical properties



**Figure S8.** Band structure and PDOS of the most stable S-substituted SL  $\text{MoO}_3$  structures for different concentrations calculated using HSE06. Labels from (a) to (k) corresponds to the structure of the Figure S5. The Fermi level is set to zero. The related structures are presented in Fig S6.



**Figure S9.** Band structure and PDOS of the most stable O-substituted SL  $\text{MoS}_2$  structures for different concentrations calculated using HSE06. Labels from (a) to (l) corresponds to the structure of the Figure S6. The Fermi level is set to zero. The related structures are presented in Fig S7.

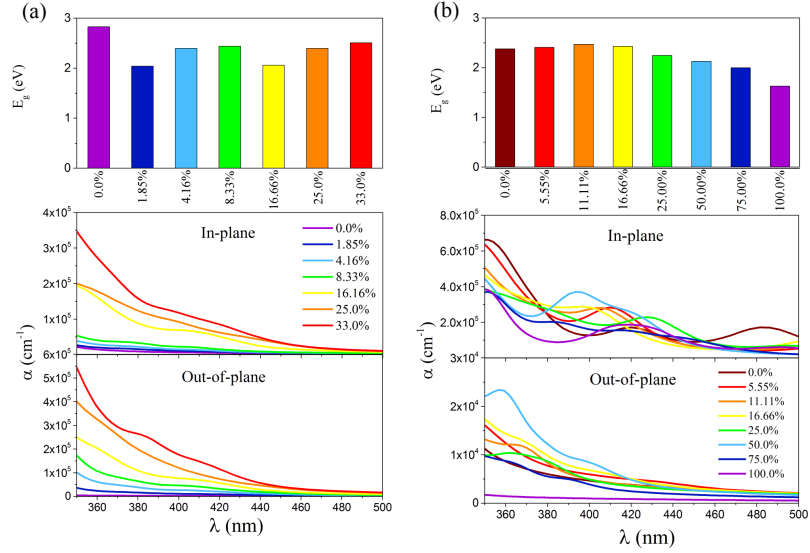


Figure S10. Computed electronic band gaps (in eV) and absorption coefficient ( $\alpha$ , in cm<sup>-1</sup>) along in-plane and out-of-plane of the most stable (a) S-doped MoO<sub>3</sub>SL and (b) O-doped MoS<sub>2</sub>SL for different S- and O-concentrations.

#### S4.4. High frequency dielectric profile $\epsilon_\infty(z)$

The four-step procedure to calculate the high frequency dielectric profile  $\epsilon_\infty(z)$  of S doped SL MoO<sub>3</sub> and O doped SL MoS<sub>2</sub> is depicted in Fig. S13a and b.

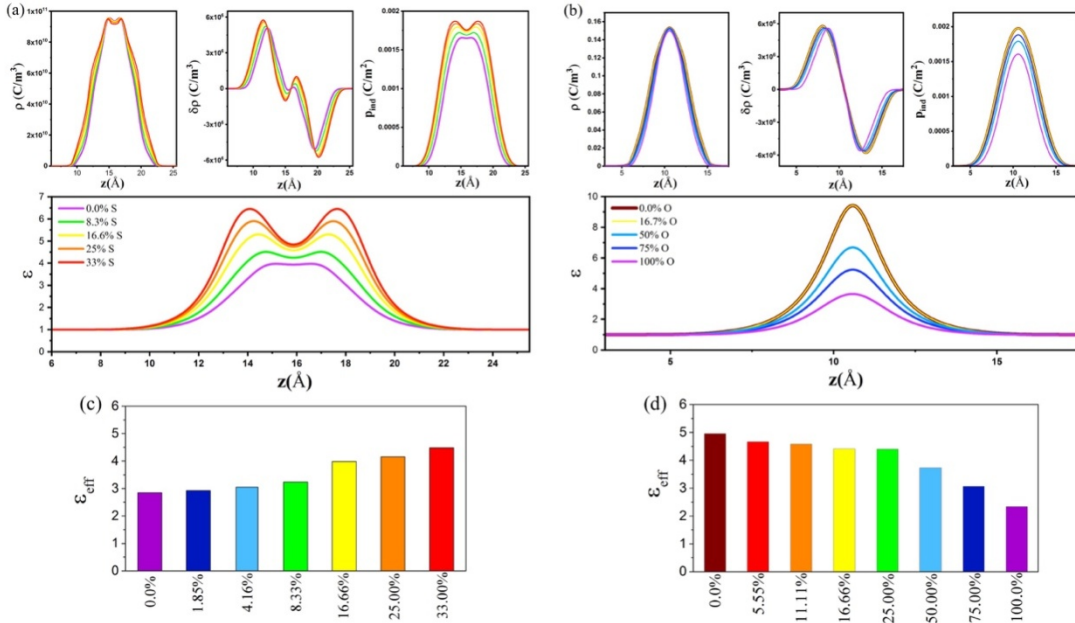


Figure S11 Four-step procedure to extract a dielectric constant profile along out-of-plane for (a) S-substituted SL MoO<sub>3</sub> and (b) O-substituted SL MoS<sub>2</sub> for different S- and O-concentrations. The effective dielectric constant along out-of-plane for (c) S-substituted SL MoO<sub>3</sub> and (d) O-substituted SL MoS<sub>2</sub> for different S- and O-concentrations.

## S4.5. Absolute valence and conduction band position

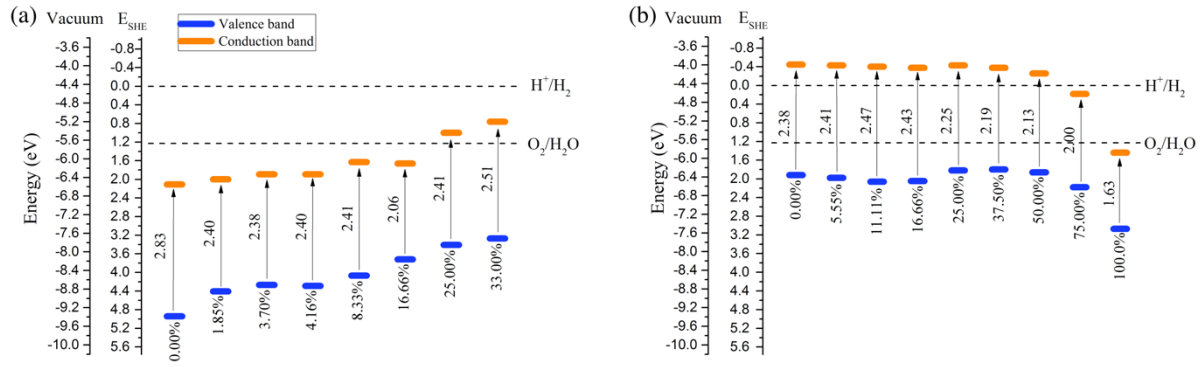


Figure S12. Calculated conduction and valence band-edge positions for (a) S-doped MoO<sub>3</sub>SL and (b) O-doped MoS<sub>2</sub>SL with respect to vacuum level and standard hydrogen electrode. The lower edge of the conduction band (orange colour) and upper edge of the valence band (blue colour) are presented along with the band gap in electron volts. The dashed black lines indicate the water stability limits for hydrogen and oxygen evolution. The absolute potential of the standard hydrogen electrode was taken as 4.44 eV at a pH = 0.

## S5. MoS<sub>x</sub>O<sub>3-x</sub>/MoS<sub>2</sub> heterojunctions

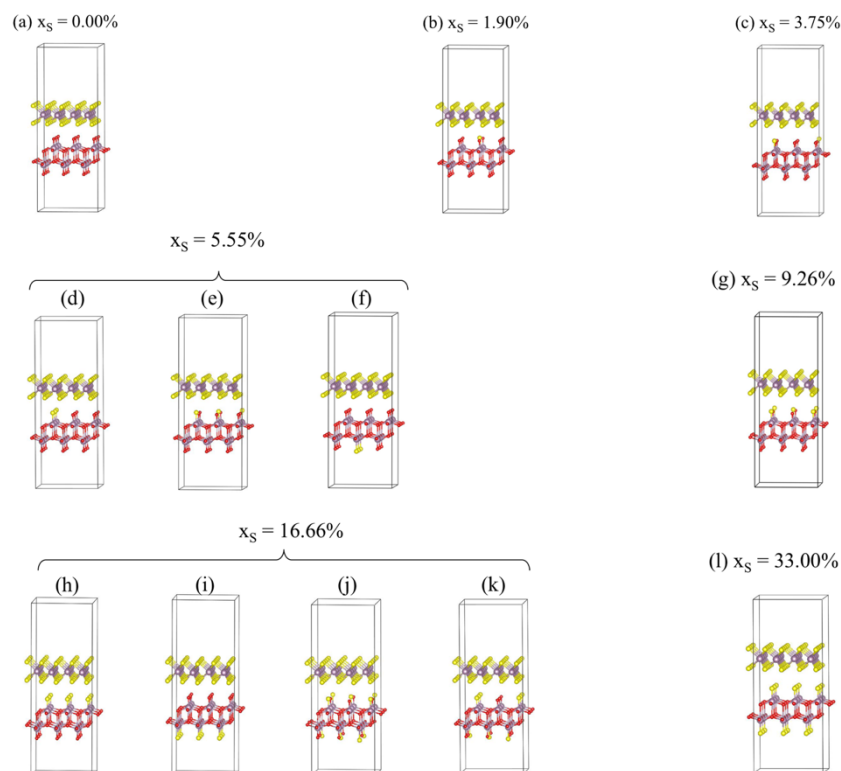
To assess the most stable configuration, the interface binding energies are calculated, which are defined as:

$$E_{b(\text{interface})} = (E_{\text{composite}} - E_{\text{MoS}_x\text{O}_{3-x}} - E_{\text{MoS}_2}) / \mathcal{A} \quad (21)$$

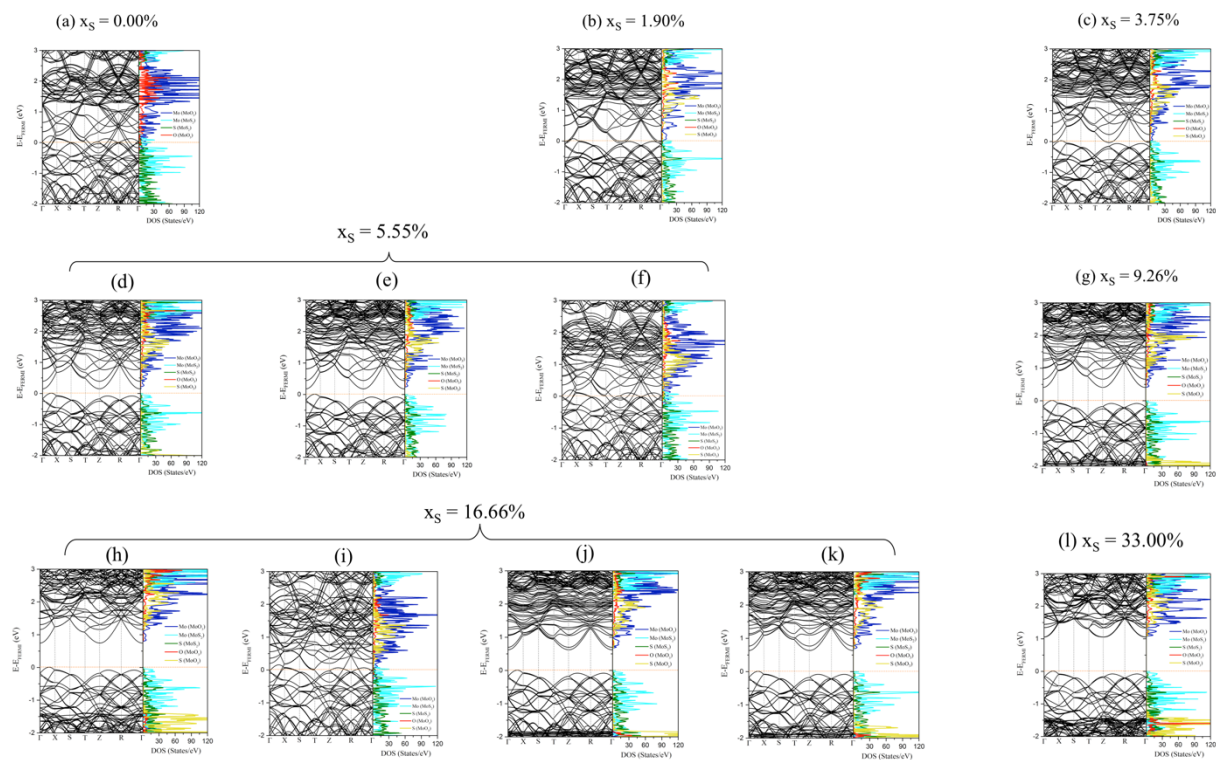
where  $E_{\text{composite}}$ ,  $E_{\text{MoS}_x\text{O}_{3-x}}$  and  $E_{\text{MoS}_2}$  are the total energies of MoS<sub>x</sub>O<sub>3-x</sub>/MoS<sub>2</sub> heterojunction, 2D MoS<sub>x</sub>O<sub>3-x</sub> layer and 2D MoS<sub>2</sub> layer, respectively. And  $\mathcal{A}$  is the in-plane area of the heterojunction unit cell.

S concentration (%) into MoO <sub>3</sub> layer	System	corresponding figure	$d$ (Å)	$E_{b(\text{interface})}$ (meV/Å <sup>2</sup> )	$E_g$ (eV)	$\epsilon_{\text{eff}}$
0.0	MoO <sub>3</sub> /MoS <sub>2</sub>	Fig. S16 (a)	2.86	-5.36	metallic	---
1.9	MoS <sub>0.06</sub> O <sub>2.94</sub> /MoS <sub>2</sub>	Fig. S16 (b)	2.96	-6.51	metallic	---
3.75	MoS <sub>0.11</sub> O <sub>2.89</sub> /MoS <sub>2</sub>	Fig. S16 (c)	3.00	-6.74	metallic	---
5.55	MoS <sub>0.17</sub> O <sub>2.83</sub> /MoS <sub>2</sub>	Fig. S16 (d)	3.04	-7.74	0.15	3.85
		Fig. S16 (e)	3.14	-7.43	0.18	3.78
		Fig. S16 (f)	2.87	-6.28	metallic	---
9.26	MoS <sub>0.28</sub> O <sub>2.72</sub> /MoS <sub>2</sub>	Fig. S16 (g)	3.18	-8.04	0.41	4.3
16.66	MoS <sub>0.5</sub> O <sub>2.5</sub> /MoS <sub>2</sub>	Fig. S16 (h)	3.21	-8.35	0.75	4.44
		Fig. S16 (i)	2.88	-5.74	Metallic	---
		Fig. S16 (j)	3.17	-7.81	0.62	4.34
		Fig. S16 (k)	3.19	-8.04	0.63	4.37
33	MoSO <sub>2</sub> /MoS <sub>2</sub>	Fig. S16 (l)	3.23	-8.65	1.05	4.52

Table S5. Different models, corresponding figures, calculated interlayer distance ( $d$ ), binding energy ( $E_b$ ), the electronic band gap ( $E_g$ ) and effective high frequency dielectric constant ( $\epsilon_{\text{eff}}$ ) along out-of-plane direction for the MoS<sub>x</sub>O<sub>3-x</sub>/MoS<sub>2</sub> for different S concentration into MoO<sub>3</sub> layer.



**Figure S13. Optimized structures of S-doped MoO<sub>3</sub>/MoS<sub>2</sub> interface for different concentrations (the increase of oxygen concentration in the MoO<sub>3</sub> layer follow the increase of the label from (a) to (l)). Color code as in Figure S1.**



**Figure S14. Band structure and PDOS of the of S-doped MoO<sub>3</sub>/MoS<sub>2</sub> interface for different concentrations calculated using HSE06. The Fermi level is set to zero. Labels from (a) to (l) corresponds to the structure of the Figure S13. The related structures are presented in Fig S16.**

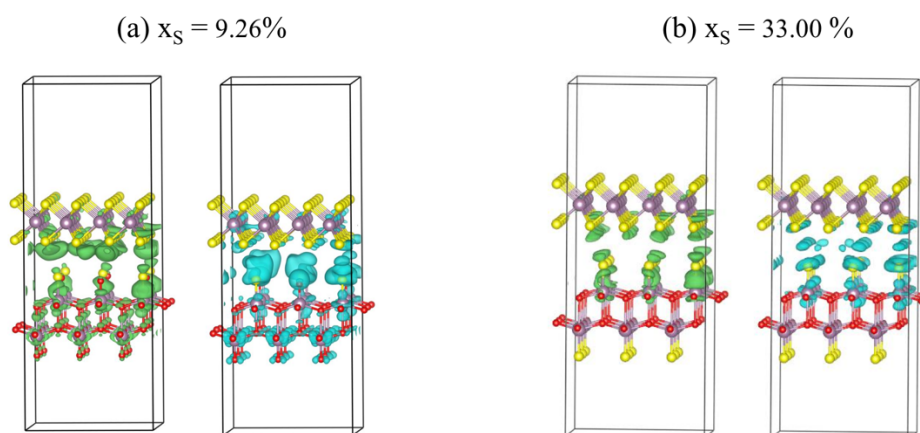


Figure S15. Charge density difference  $\Delta\rho = \rho(\text{MoO}_{3-x}\text{S}_x/\text{MoS}_2) - \rho(\text{MoO}_{3-x}\text{S}_x) - \rho(\text{MoS}_2)$  for  $\text{MoO}_{3-x}\text{S}_x/\text{MoS}_2$  heterojunction with  $x_S = 9.26\%$  (a) and  $x_S = 33.00\%$  (b).  $\Delta\rho$  is shown by isosurfaces bounding regions of electron accumulation at  $+1.2 \times 10^{-4} \text{ e}/\text{\AA}^3$  (cyan) and electron depletion at  $-1.2 \times 10^{-4} \text{ e}/\text{\AA}^3$  (green).

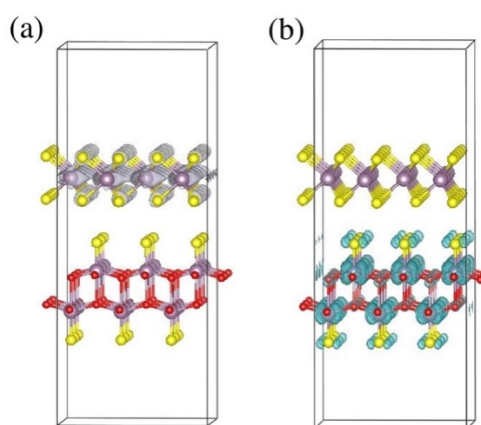


Figure S16. The partial charge densities of the VBM (a) and the CBM (b) for  $\text{MoSO}_2/\text{MoS}_2$  interface.

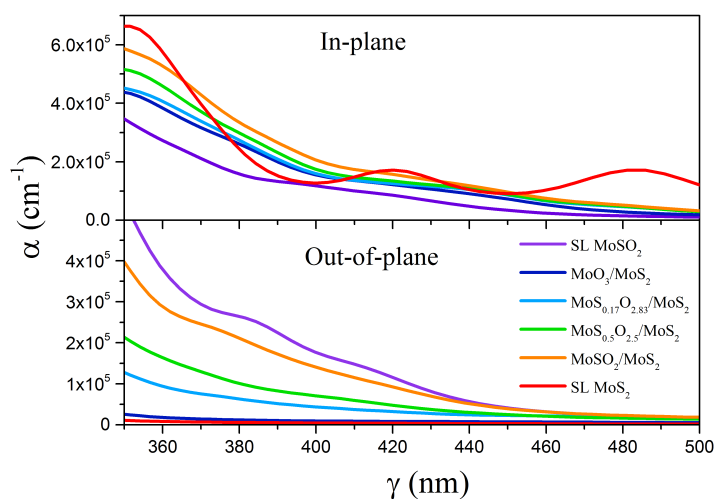


Figure S17. Absorption coefficient ( $\alpha$ , in  $\text{cm}^{-1}$ ) along in-plane and out-of-plane for SL  $\text{MoSO}_2$ , SL  $\text{MoS}_2$ ,  $\text{MoO}_3/\text{MoS}_2$  and  $\text{MoS}_x\text{O}_{3-x}/\text{MoS}_2$  structures.

## S6. Impact of increasing layers in $\text{MoS}_x\text{O}_{3-x}/\text{MoS}_2$ heterostructures

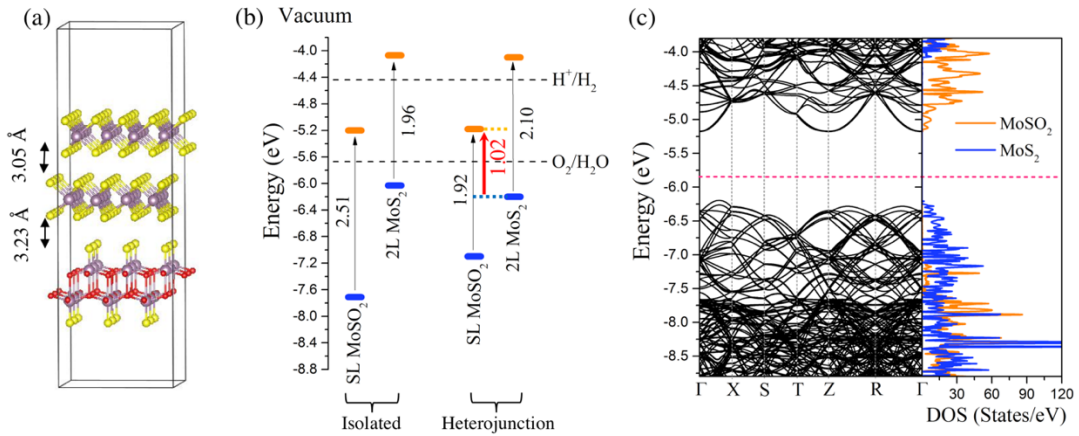


Figure S18. Optimized structure (a), conduction and valence band-edge positions respect to vacuum level (b), band structure and PDOS (c) for each layer for  $\text{MoSO}_2/2\text{L-MoS}_2$  heterostructure.

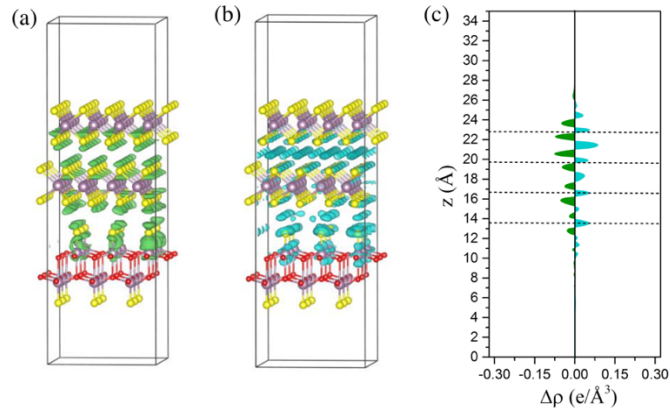


Figure S19. Charge density difference  $\Delta\rho$  (a and b) and planar-averaged charge density difference  $\langle\Delta\rho(z)\rangle$  (c) for  $\text{MoSO}_2/2\text{L-MoS}_2$  heterostructure.  $\Delta\rho$  is shown by isosurfaces bounding regions of electron accumulation at  $+1.2 \times 10^{-4} \text{ e}/\text{\AA}^3$  (cyan) and electron depletion at  $-1.2 \times 10^{-4} \text{ e}/\text{\AA}^3$  (green).

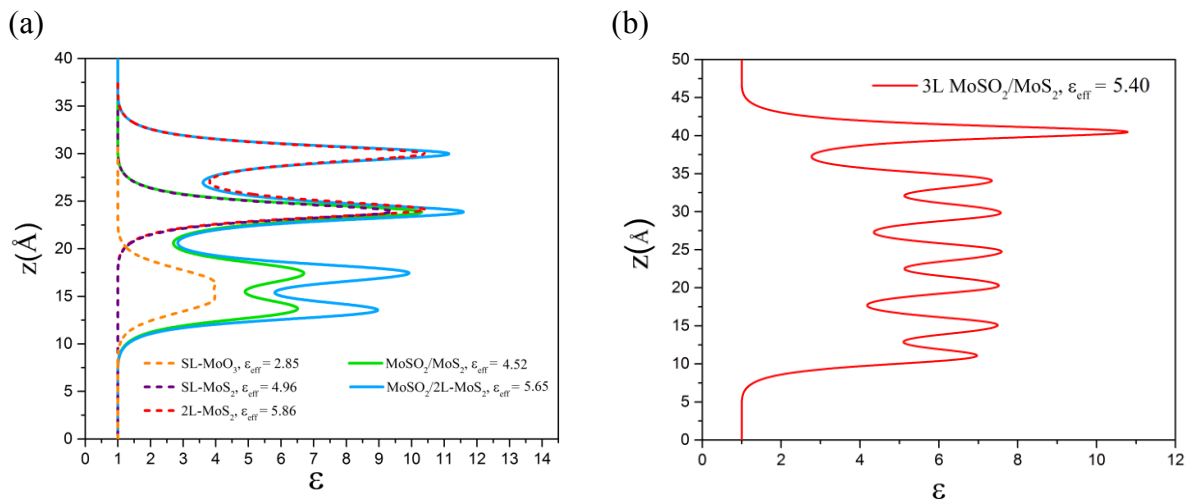


Figure S20. (a) Dielectric constant profile along out-of-plane for SL  $\text{MoO}_3$ , SL  $\text{MoS}_2$ , 2L  $\text{MoS}_2$ ,  $\text{MoSO}_2/\text{MoS}_2$  and  $\text{MoSO}_2/2\text{L-MoS}_2$  heterostructures. The effective dielectric constant for each structure is also reported in the legend. (b) Dielectric constant profile along out-of-plane for 3L- $\text{MoSO}_2/\text{MoS}_2$  structures. The effective dielectric constant for each structure is also reported in this figure.



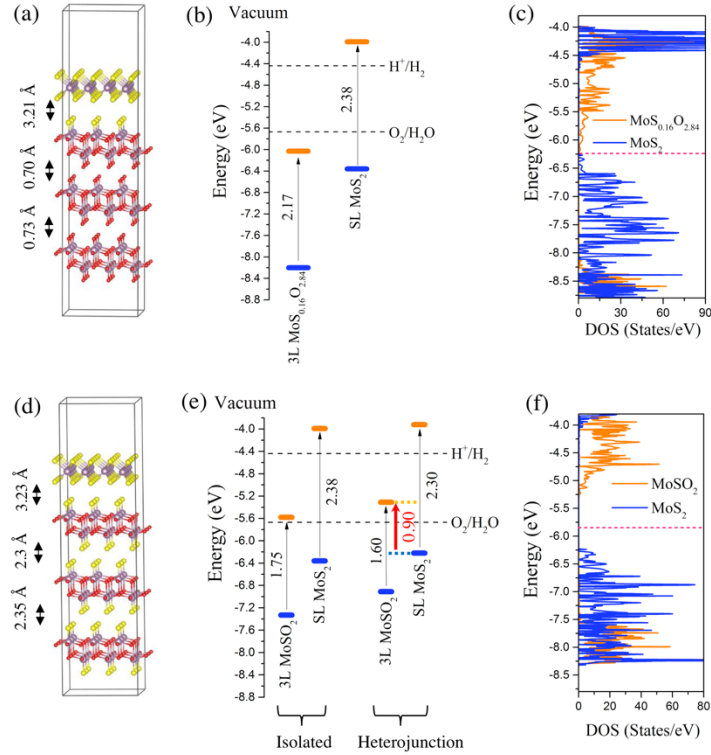


Figure S21. Optimized structure, conduction and valence band-edge positions respect to vacuum level and PDOS for each layer for 3L-MoS<sub>0.17</sub>O<sub>2.83</sub>/MoS<sub>2</sub> (a-c) and 3L-MoSO<sub>2</sub>/MoS<sub>2</sub> (d-f) heterostructures.

## S7. Atomic structures for VASP POSCAR file

### S7.1. Atomic structure of S-doped MoO<sub>3</sub>/MoS<sub>2</sub> in VASP POSCAR

MoO<sub>3</sub>/MoS<sub>2</sub>

1.0

```

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-0.0000917879   11.1179943081    0.0000000000
0.0000000000    0.0000000000    35.0000000000

```

Mo O S

34 54 32

Direct

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0.999968827    0.331849605    0.502099991
0.875089109    0.082052000    0.502938807
0.374872506    0.081618406    0.502990901
0.124897622    0.082158603    0.502937257
0.749927819    0.331847668    0.502068162
0.144783393    0.916487038    0.326637238
0.478390664    0.916491032    0.326831460
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0.144556403    0.583125651    0.326904714
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0.144586489    0.250097662    0.326960534
0.811687112    0.250225067    0.326888233
0.810407043    0.916471362    0.326634049
0.856486738    0.749815822    0.240421519

```

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0.523140728	0.750036240	0.240457997
0.999966383	0.831517220	0.503379524
0.250007898	0.831866145	0.503152192
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0.375012189	0.581947803	0.502245367
0.856516838	0.083355673	0.240457773
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0.327060848	0.916501880	0.320242584
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0.488690495	0.577651262	0.375298142
0.824328303	0.581244409	0.375389516
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0.489981085	0.915991426	0.375409991
0.817047775	0.916005909	0.375257552
0.993158817	0.583345950	0.320058316
0.153917372	0.250875860	0.375596672
0.659951329	0.583008230	0.320864111
0.511773050	0.082907803	0.192040801
0.326068968	0.249939740	0.320667744
0.006868987	0.749734759	0.247246057
0.511089504	0.416368783	0.191793367
0.326032877	0.583273470	0.320622176
0.845266700	0.083616398	0.192009673
0.178400770	0.083574772	0.192058384
0.673554659	0.750156045	0.247267127
0.340219051	0.749879718	0.247276574
0.673493028	0.416491687	0.246904641
0.340070128	0.416504741	0.247118846
0.006900989	0.416516066	0.247369960
0.006891152	0.083457604	0.247323036
0.673578918	0.083047137	0.247298047
0.340229601	0.083334975	0.247323766
0.511740804	0.750448346	0.192015454
0.178410187	0.749899149	0.192006990
0.845108807	0.749761343	0.191975161
0.845716417	0.416323990	0.191949517
0.166233212	0.249952570	0.256444752
0.499628335	0.249797344	0.256302416
0.177889854	0.416272014	0.191989884

0.832645357	0.583172143	0.256317496
0.833318472	0.916587532	0.256291628
0.832675159	0.249925122	0.256404668
0.499230951	0.916603565	0.256458312
0.498961180	0.083785631	0.311155260
0.832252085	0.083298251	0.310889721
0.165642723	0.416582316	0.310790360
0.166071832	0.916598976	0.256387800
0.498747170	0.416615814	0.309960902
0.498970628	0.749318898	0.311003566
0.832249463	0.749791324	0.310746223
0.499643147	0.583315372	0.256215602
0.165507004	0.749833703	0.310626537
0.659963250	0.250171781	0.320856333
0.993161261	0.249849230	0.320192814
0.166271374	0.583168507	0.256324977
0.832371294	0.416568220	0.310467988
0.165504888	0.083296172	0.310775518
0.375085324	0.917624235	0.548913062
0.999970376	0.168900847	0.548486590
0.499973387	0.168701306	0.548474252
0.749988854	0.168785945	0.548477829
0.125027865	0.417514712	0.547849596
0.375048071	0.417426616	0.547853649
0.624935031	0.417401850	0.547855794
0.874910116	0.417503238	0.547849119
0.999969900	0.665857553	0.548397362
0.499978870	0.666855991	0.548256159
0.749963701	0.666511834	0.548295736
0.125100732	0.917968035	0.549021363
0.249984711	0.168821752	0.548480690
0.250003844	0.666331172	0.548321486
0.124363825	0.917457044	0.457820505
0.625398755	0.917579710	0.457115293
0.624914110	0.917546988	0.548899174
0.999932706	0.166466564	0.456689000
0.250308782	0.166178361	0.456849247
0.500058472	0.165760800	0.457089931
0.749616861	0.166054204	0.456927836
0.124934852	0.417814076	0.456265897
0.875802577	0.917451918	0.457689226
0.374927551	0.417705715	0.456137598
0.875046313	0.417781055	0.456235021
0.999944627	0.668909431	0.456592292
0.250226676	0.669073462	0.456694931
0.500040770	0.669508517	0.456888497
0.749705493	0.669187486	0.456754506
0.374342501	0.917544723	0.457217187
0.625013351	0.417696923	0.456113935
0.874809623	0.917918503	0.548998177

## S7.2. Atomic structure of S-doped MoSO<sub>2</sub>/MoS<sub>2</sub> in VASP POSCAR

MoSO<sub>2</sub>/MoS<sub>2</sub>

1.0

11.6662597656	0.0000000000	0.0000000000
-0.0000461254	11.1740646361	0.0000000000
0.0000000000	0.0000000000	35.0000000000

Mo O S

34 36 50

Direct

0.808510721	0.916379809	0.325317770
0.475986212	0.916414440	0.325271070
0.142480209	0.916400611	0.325266153
0.475399524	0.582808912	0.325273126
0.142191634	0.582985997	0.325345099
0.809067726	0.582939982	0.325369954
0.475510150	0.250089824	0.325276792
0.142255917	0.249862000	0.325346470
0.809113443	0.249868155	0.325374454
0.857672870	0.749753535	0.240043283
0.190990135	0.749709964	0.239993975
0.524378419	0.749785721	0.239977837
0.190995872	0.416431516	0.239956304
0.857737124	0.416421384	0.239986062
0.191027269	0.083129004	0.240001976
0.857709885	0.083072759	0.240050241
0.524421751	0.083071925	0.239987209
0.374845386	0.081813663	0.524458170
0.124849215	0.082211226	0.524426460
0.875027359	0.082131177	0.524436712
0.250001192	0.331986934	0.523752630
0.624989331	0.081735320	0.524473846
0.999923885	0.331955731	0.523770690
0.749882579	0.832050443	0.524512291
0.499925107	0.832273841	0.524420559
0.249970198	0.831927836	0.524569035
0.999927163	0.831672788	0.524682939
0.874931097	0.582034886	0.523819566
0.624936163	0.581963480	0.523874104
0.374941319	0.581961393	0.523860216
0.124912061	0.582034051	0.523808837
0.749855697	0.331977308	0.523742855
0.499938726	0.332003504	0.523722231
0.524374723	0.416437894	0.239868358
0.990544021	0.916545033	0.320083052
0.009257318	0.749684215	0.246035412
0.166570038	0.583050311	0.253950536
0.833112597	0.583018064	0.254038721
0.499963045	0.249778539	0.253940940
0.166595578	0.249773934	0.253976047
0.833164930	0.249794632	0.254040927
0.675947785	0.749846518	0.246027187
0.009294863	0.083043396	0.246047705

0.009294075	0.416541040	0.246009722
0.675919294	0.416538566	0.245882720
0.342607081	0.416544646	0.245882362
0.499951869	0.583055139	0.253917783
0.675998509	0.082896940	0.246037409
0.342623115	0.083030000	0.245939821
0.342585653	0.749711931	0.245932981
0.499758601	0.916459560	0.253919691
0.166566610	0.916447401	0.254049152
0.323802322	0.583017111	0.319435954
0.657376647	0.582973778	0.319749802
0.990716219	0.582963228	0.319259435
0.324396819	0.916590273	0.319337904
0.657158315	0.916544914	0.318767250
0.833489776	0.916440070	0.253973305
0.657449722	0.249743402	0.319688290
0.990732014	0.249754161	0.319348067
0.500021279	0.749555707	0.311300129
0.166665524	0.749700189	0.311257273
0.323893070	0.249687508	0.319474906
0.500157475	0.416468292	0.311032832
0.166843638	0.416425496	0.311232060
0.833391905	0.416430980	0.311185449
0.500073969	0.083280317	0.311355740
0.166637048	0.083132587	0.311292976
0.833447278	0.083052784	0.311430156
0.833413661	0.749736845	0.311362147
0.177763313	0.083007708	0.179327533
0.490301788	0.915843785	0.385891616
0.159482524	0.915898621	0.385772109
0.817399144	0.915755749	0.386004299
0.488240570	0.578738391	0.385935247
0.154455319	0.582106888	0.386040926
0.824322760	0.580912888	0.385994762
0.488656402	0.254794598	0.385925233
0.154807776	0.251175016	0.386036366
0.823847830	0.252417862	0.386011660
0.844715178	0.749832869	0.179359838
0.511652410	0.749928594	0.179292604
0.177714080	0.749855936	0.179322377
0.844960093	0.416406453	0.179296792
0.511347473	0.416416645	0.179185852
0.177696496	0.416414917	0.179277286
0.844793141	0.083016634	0.179364458
0.511716068	0.082961999	0.179299697
0.749931812	0.168804899	0.569973290
0.749719977	0.669101000	0.478397518
0.499978364	0.669251442	0.478491217
0.250136167	0.668995559	0.478338778
0.999893010	0.668882549	0.478227019
0.874987483	0.417920768	0.477996677
0.624968588	0.417901188	0.477868319
0.374883503	0.417914331	0.477898180

0.124884181	0.417945653	0.478019446
0.749716163	0.166716784	0.478571206
0.499976188	0.166534767	0.478675187
0.250131220	0.166826129	0.478501409
0.999887288	0.167031214	0.478368968
0.124585815	0.917886019	0.479210556
0.374503791	0.917936683	0.478797525
0.625201941	0.917965472	0.478715301
0.875413656	0.917898595	0.479118556
0.874843836	0.918014944	0.570333838
0.374998838	0.917871892	0.570294738
0.124983951	0.918055832	0.570346832
0.749931216	0.667001426	0.569800436
0.499927461	0.667218447	0.569769502
0.249929652	0.666864455	0.569813967
0.999929249	0.666605830	0.569857061
0.874874353	0.417843223	0.569465458
0.624896049	0.417768002	0.569478810
0.374986678	0.417784393	0.569479287
0.124964103	0.417859674	0.569466233
0.499921083	0.168741301	0.569959402
0.249930874	0.168839678	0.569970191
0.999932468	0.168916538	0.569987297
0.624878466	0.917822778	0.570285439

### S7.3. Atomic structure of S-doped MoSO<sub>2</sub>/2L MoS<sub>2</sub> in VASP POSCAR

MoSO<sub>2</sub>/2L MoS<sub>2</sub>

1.0

11.6662597656	0.0000000000	0.0000000000
-0.0000461254	11.1740646361	0.0000000000
0.0000000000	0.0000000000	42.0000000000

Mo O S

50 36 82

Direct

0.808429062	0.916536570	0.269409746
0.475938261	0.916570902	0.269256145
0.142526180	0.916553020	0.269346029
0.475275040	0.582972705	0.269291341
0.142096564	0.583160341	0.269353092
0.808962822	0.583133876	0.269437075
0.475402027	0.250290781	0.269283950
0.142186224	0.250070691	0.269355059
0.809043050	0.250060201	0.269436598
0.857838750	0.749932826	0.198344305
0.191154569	0.749914169	0.198257670
0.524572134	0.749943852	0.198215052
0.191135749	0.416604728	0.198209509
0.857876539	0.416599214	0.198293656
0.191190869	0.083268322	0.198276341
0.857875705	0.083234355	0.198363662
0.524615884	0.083260968	0.198228925
0.374827594	0.083668776	0.435031384

0.124826670	0.084056281	0.434999049
0.874987185	0.083982803	0.435007334
0.249961004	0.333839864	0.434557229
0.624953985	0.083594233	0.435045481
0.999897003	0.333801597	0.434585482
0.749857843	0.833896458	0.435044825
0.499897301	0.834104061	0.434977889
0.249936894	0.833785594	0.435086399
0.999901056	0.833541751	0.435174763
0.874902248	0.583893239	0.434581906
0.624905944	0.583811164	0.434620231
0.374911785	0.583812773	0.434609473
0.124884576	0.583893776	0.434573501
0.749833524	0.333837718	0.434540480
0.499905914	0.333868265	0.434510052
0.749947369	0.666248024	0.583629608
0.499950141	0.666244268	0.583628476
0.249952212	0.666248858	0.583630621
0.999949336	0.666252494	0.583631337
0.874951184	0.416264951	0.583638549
0.624951243	0.416264832	0.583640635
0.374948144	0.416264921	0.583639503
0.124948867	0.416264921	0.583637357
0.749947846	0.166273534	0.583630264
0.499950260	0.166272625	0.583629489
0.249951810	0.166275322	0.583630860
0.999949336	0.166276544	0.583631814
0.374949574	0.916258574	0.583622098
0.874949694	0.916257143	0.583620965
0.624949336	0.916258991	0.583621442
0.124950096	0.916256607	0.583621502
0.524558246	0.416620076	0.198131040
0.990533769	0.916707695	0.265180856
0.009405996	0.749790967	0.203359634
0.166724786	0.583232462	0.209853262
0.833198547	0.583192766	0.210000247
0.500103831	0.249966547	0.209848315
0.166748613	0.249937564	0.209882215
0.833262086	0.249974445	0.210007414
0.676092029	0.750028014	0.203339353
0.009443047	0.083267108	0.203383967
0.009426985	0.416718900	0.203332394
0.676064551	0.416704029	0.203219727
0.342741728	0.416712523	0.203151301
0.500094473	0.583220899	0.209831372
0.676141500	0.083050057	0.203357056
0.342786551	0.083196290	0.203220293
0.342744172	0.749879122	0.203205660
0.499874085	0.916623950	0.209783465
0.166623890	0.916613698	0.210008889
0.323714048	0.583195746	0.264362395
0.657286704	0.583213687	0.264749259
0.990628183	0.583128393	0.264282525

0.324395359	0.916733146	0.264231056
0.657124639	0.916716218	0.263843536
0.833719432	0.916600943	0.209976286
0.657393575	0.249879763	0.264682531
0.990673482	0.249948680	0.264332473
0.499982148	0.749688327	0.257634938
0.166596815	0.749884069	0.257629484
0.323817492	0.249879405	0.264379919
0.500128984	0.416659862	0.257424682
0.166794270	0.416610897	0.257585377
0.833270550	0.416628957	0.257613778
0.500019610	0.083478928	0.257691294
0.166583478	0.083287992	0.257683486
0.833423257	0.083214648	0.257837147
0.833398700	0.749914169	0.257767141
0.177796081	0.083082899	0.147718161
0.489970863	0.915849209	0.319774270
0.160444722	0.915756345	0.319747776
0.816849411	0.915646553	0.319986582
0.487840831	0.578801751	0.319853544
0.154502854	0.582279921	0.319934577
0.824334621	0.580959082	0.319955409
0.488258392	0.255262643	0.319830090
0.154813543	0.251622856	0.319930136
0.823978841	0.252998233	0.319957107
0.844867229	0.750172257	0.147775859
0.512104213	0.750197649	0.147636607
0.177776635	0.750174344	0.147698104
0.845129311	0.416527152	0.147718310
0.511740565	0.416573405	0.147562817
0.177741379	0.416537255	0.147645280
0.844949126	0.083064929	0.147793174
0.512186170	0.083087936	0.147649065
0.249946088	0.330350101	0.621748745
0.999951839	0.330353439	0.621748745
0.874948084	0.080343492	0.621738732
0.624948680	0.080343641	0.621738315
0.499948323	0.330343634	0.621749759
0.749953866	0.330346256	0.621749640
0.124950580	0.580363393	0.621745825
0.374951869	0.080345489	0.621738911
0.374952048	0.580366671	0.621745825
0.124951273	0.080344997	0.621739089
0.624949038	0.580369115	0.621745706
0.999948740	0.830366194	0.621736169
0.874948204	0.580366075	0.621745944
0.749948025	0.830362797	0.621734917
0.499951303	0.830359042	0.621734619
0.249951392	0.830363035	0.621736109
0.749903619	0.170547292	0.472915560
0.499948502	0.830242932	0.545579791
0.749690473	0.670765460	0.396703035
0.499951124	0.670926213	0.396782249



0.250106096	0.670657456	0.396651506
0.999864936	0.670535326	0.396556526
0.875008583	0.419731617	0.396450460
0.624966800	0.419736356	0.396318585
0.374801844	0.419740498	0.396349043
0.124823429	0.419748694	0.396472424
0.749661565	0.168780521	0.396840870
0.499951631	0.168577373	0.396925151
0.250124574	0.168894336	0.396785647
0.999856651	0.169116423	0.396677196
0.124562107	0.919822752	0.397297710
0.374499410	0.919840336	0.396965623
0.625158250	0.919861257	0.396898508
0.875372291	0.919826448	0.397225022
0.249951407	0.830241084	0.545578897
0.999949813	0.830240786	0.545577347
0.874950647	0.580250204	0.545584857
0.624950707	0.580241084	0.545581758
0.374947250	0.580244303	0.545582294
0.124949396	0.580253720	0.545585394
0.749946058	0.330310732	0.545587957
0.499951482	0.330311507	0.545589089
0.249953181	0.330309242	0.545586288
0.999947429	0.330308944	0.545585334
0.874951661	0.080307111	0.545583427
0.624949932	0.080304123	0.545579791
0.374947995	0.080303460	0.545581102
0.124948330	0.080306955	0.545584559
0.749947608	0.830241382	0.545578718
0.874828756	0.919915438	0.473137319
0.374957889	0.919773877	0.473112345
0.124947734	0.919949710	0.473147154
0.749896705	0.669143915	0.472788006
0.499899209	0.669318855	0.472768724
0.249907017	0.669033647	0.472794622
0.999900520	0.668813884	0.472824872
0.874821842	0.419831514	0.472591549
0.624850214	0.419727951	0.472579092
0.374978334	0.419752061	0.472584218
0.124948360	0.419855833	0.472597480
0.499893546	0.170526847	0.472903013
0.249901652	0.170551777	0.472915739
0.999900877	0.170581877	0.472931057
0.624862909	0.919729888	0.473105699

#### S7.4. Atomic structure of 3L S-doped MoSO<sub>2</sub>/MoS<sub>2</sub> in VASP POSCAR

3L S-doped MoSO<sub>2</sub>/MoS<sub>2</sub>

1.0

11.1740646362	0.0000000000	0.0000000000
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0.0000000000	0.0000000000	52.0000000000

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## Direct

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